

## Preparation of Nanocrystalline $WO_3$ and $MoO_3$ by Different Sol–Gel Methods

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### Abstract

Two sol–gel methods for the preparation of  $WO_3$  and  $MoO_3$  nanopowders were used in this work: i) an ion-exchange reaction and ii) an oxidizing reaction ( $M + H_2O_2$ ). The phase and structural transformations undergone by colloidal solutions of tungsten acid (i) and peroxotungsten and peroxomolybdic acids (ii) as a function of thermal treatment were investigated by X-ray diffraction (XRD) and infrared spectroscopy (IR). Depending on the methods used, different phases were obtained: crystalline hydrates, amorphous and nanocrystalline products. The tungsten trioxide hydrates were prepared by the ion exchange method, crystallization in  $m$ - $WO_3$  occurring above 300 °C. The tungsten sample being formed in the oxidizing reaction remained amorphous up to 300 °C; above 300 °C,  $m$ - $WO_3$  crystallized. The particle size of  $m$ - $WO_3$  was 15 nm irrespective of the methods applied. IR analysis showed that amorphous tungsten network was built by distorted  $WO_6$  units without participation of peroxo groups ( $O_2^{2-}$ ). The preparation of  $MoO_3$  nanopowders by an oxidizing reaction was also studied. Crystallization of  $MoO_3$  was found to start earlier (200 °C), leading to completely crystallized  $o$ - $MoO_3$  at 300 °C. The amorphous state of the product was detected at 100 °C only. Comparative analysis of the methods applied showed the oxidizing method to be more suitable for obtaining nanoparticles.

### INTRODUCTION

$WO_3$  and  $MoO_3$  are well known metal oxide materials used in gas sensing, catalytic, photochromic and electrochromic research fields. Depending on temperature, there are several crystalline modifications of  $WO_3$  (triclinic, monoclinic, orthorhombic, tetragonal and hexagonal) [1, 2] while  $MoO_3$  can exist in two polymorphous forms: orthorhombic and monoclinic [3]. All polymorphs of  $WO_3$  can be described as distortions from the cubic  $ReO_3$  structure, which consists of a three-dimensional network of corner-sharing  $WO_6$  octahedra. The  $ReO_3$  structure is characteristic of  $m$ - $MoO_3$  only. The monoclinic  $MoO_3$  is transformed into orthorhombic  $MoO_3$  at 400 °C. Crystalline  $o$ - $MoO_3$  has a layered structure of distorted  $MoO_6$  polyhedra sharing both edges and corners. One oxygen in each polyhedron is unshared  $M=O$  [4]. From the practical point of view, the production of  $MoO_3$  and  $WO_3$  nanopowders is interesting. Several

methods have been applied up to now: evaporation [5], sputtering [6] and sol–gel technology [7–14]. There are different sol–gel routes based on different starting compounds: polymerization and polycondensation of metal alkoxides [7], ion exchange methods [8–10] and oxidizing reaction of metallic W or Mo with a solution of  $H_2O_2$  [11–13].

The purpose of this study is to obtain  $WO_3$  and  $MoO_3$  nanopowders using different sol–gel methods and to compare phase and structure transformations undergone by the precursors.

### EXPERIMENTAL

Two different methods were selected in this study.

i) Ion exchange method (A). An aqueous solution of  $Na_2WO_4$  was passed through ion-exchange resin to give the sol of tungsten acid [8]. The yellow sol obtained without any organic additives underwent gelation after 30 min at room temperature.

ii) Oxidizing reaction (B) between metal (W or Mo) powder and hydrogen peroxide solution ( $M + H_2O_2$ ) [11]. The sol of peroxotungsten acid was obtained in the presence of ethanol and glacial acetic acid. The peroxotungsten acid was dried at 100 °C to remove the excess of hydrogen peroxide. The obtained sol underwent gelation after 48 h at room temperature.

MoO<sub>3</sub> was prepared by oxidizing reaction (B). The sol of peroxomolybdic acid was obtained without organic additives. The obtained sol was transformed into a gel at room temperature too slowly (after two weeks).

The samples obtained were investigated by X-ray phase analysis (APT-15 Philips diffractometer, CuK<sub>α</sub> radiation) and infrared spectroscopy in the 1200–400 cm<sup>-1</sup> region using the KBr pellet technique (Nicolet-320 FTIR spectrometer). The crystallite sizes were calculated using the Scherrer method for the (222) diffraction peak of monoclinic *m*-WO<sub>3</sub> and (111) diffraction peak of orthorhombic *o*-MoO<sub>3</sub> powders (the error of crystallite size determination is ±0.3 nm).

## RESULTS AND DISCUSSION

### X-ray diffraction analysis

X-ray diffraction patterns of the tungsten sample obtained by the ion exchange method and heated at 100–300 °C are shown in Fig. 1, *a*. There are principal peaks for WO<sub>3</sub>·2H<sub>2</sub>O ( $d = 3.67$  Å,  $d = 3.21$  Å,  $d = 2.58$  Å), (18-1419 JCPDS) and for WO<sub>3</sub>·0.33H<sub>2</sub>O ( $d = 4.90$  Å,  $d = 3.84$  Å,  $d = 3.15$  Å), (35-0270 JCPDS). The ratio between the amorphous and crystalline phases is 1 : 1 (error 10 %). The thermal treatment above 300 °C leads to a visible improvement of the sample crystallinity which is indicated by a strong increase in the intensity of the peaks of *m*-WO<sub>3</sub> (83-0951 JCPDS) (see Fig. 1, *b*). The XRD pattern of the sample heated at 400 °C shows a trace of WO<sub>3</sub>·2H<sub>2</sub>O. Similar results on the presence of hydrated tungsten oxide were obtained using Raman spectroscopy in [10]. The calculated crystallite size of the *m*-WO<sub>3</sub> powder is 14.5 nm at 400 °C and increases up to 16.7 nm at 600 °C.

The tungsten sample prepared by W + H<sub>2</sub>O<sub>2</sub> reaction is amorphous up to 300 °C (Fig. 2). Above

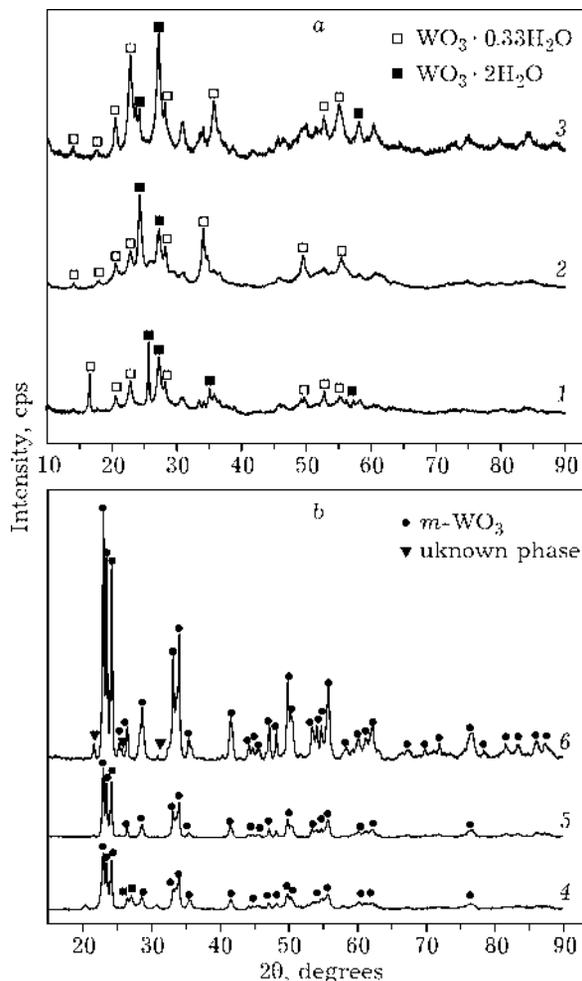


Fig. 1. XRD patterns of the tungsten sample obtained by method A. Temperature, °C: 100 (1), 200 (2), 300 (3), 400 (4), 500 (5), 600 (6).

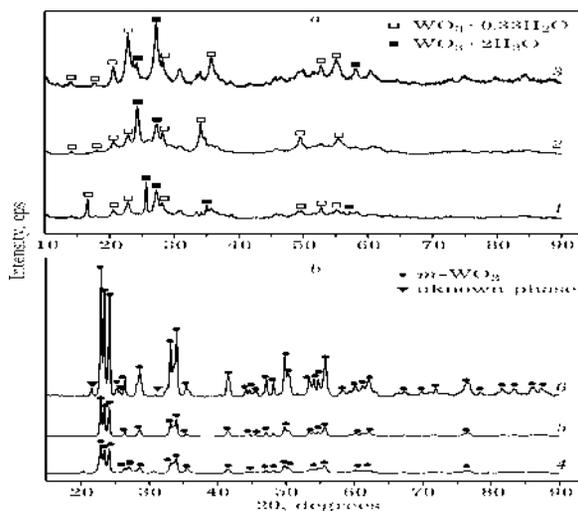


Fig. 2. XRD patterns of the tungsten sample prepared by method B. Temperature, °C: 100 (1), 200 (2), 300 (3), 400 (4).



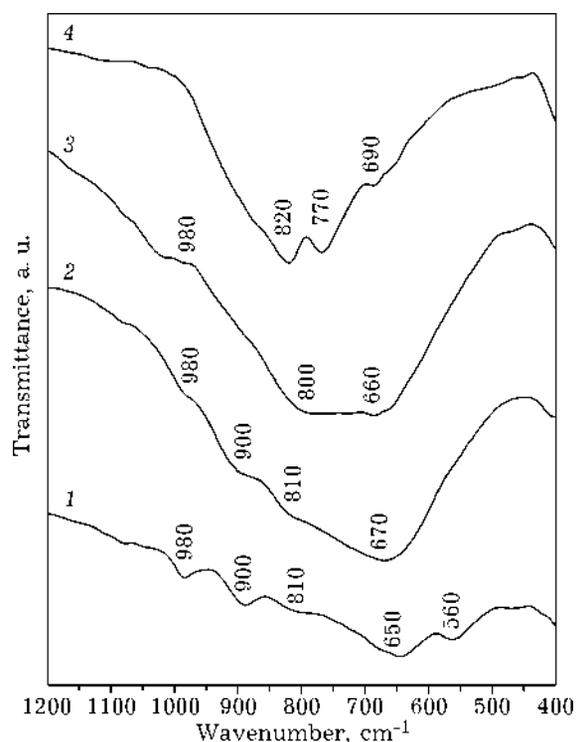


Fig. 5. IR spectra of the tungsten sample prepared by method B. Temperature, °C: 100 (1), 200 (2), 300 (3), 400 (4).

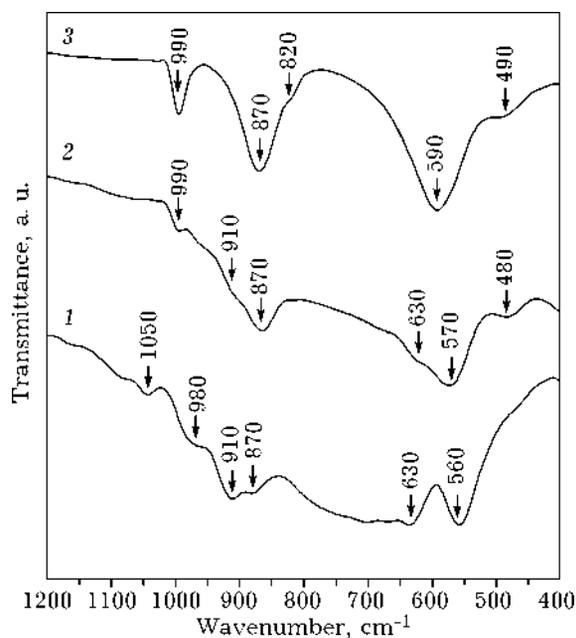


Fig. 6. IR spectra of the molybdenum sample obtained by method B. Temperature, °C: 100 (1), 200 (2), 300 (3).

signed to the vibration of peroxo (W–O–O) groups disappears at 200 °C while the band at 900  $\text{cm}^{-1}$  attributed to the vibration of (O–O) bond vanishes at 300 °C. The weak band at 980  $\text{cm}^{-1}$  and bands in the broad absorption region between 800 and 650  $\text{cm}^{-1}$  in the spectrum at 300 °C can be assigned to stretching vibrations of distorted  $\text{WO}_6$ , building the amorphous network [11]. The presence of bands characteristics of  $\text{WO}_6$  units is an indication that the number of peroxo groups ( $\text{O}_2^{2-}$ ) decreases in the amorphous network formed at 300 °C. Bands typical of crystalline  $m\text{-WO}_3$  appear in the spectrum of the sample thermally treated at 400 °C.

Figure 6 presents infrared spectra of the sample prepared by the  $\text{Mo} + \text{H}_2\text{O}_2$  reaction. The spectra of the sample heated at 100 °C show bands characteristic of peroxo complexes: 980  $\text{cm}^{-1}$  ( $\nu\text{Mo}=\text{O}$ ), 910  $\text{cm}^{-1}$  ( $\nu\text{O}-\text{O}$ ), 870  $\text{cm}^{-1}$  ( $\nu\text{Mo}-\text{O}-\text{Mo}$ ), 630  $\text{cm}^{-1}$  ( $\nu_s\text{Mo}-\text{O}-\text{O}$ ) and 560  $\text{cm}^{-1}$  ( $\nu_{as}\text{Mo}-\text{O}-\text{O}$ ) [14, 15]. The bands at 910 and 630  $\text{cm}^{-1}$  characteristic of peroxo groups are transformed into shoulders with increasing temperature (200 °C) and disappear at 300 °C. The infrared spectrum of the sample thermally treated at 300 °C is typical of  $o\text{-MoO}_3$ .

## CONCLUSIONS

1. Depending on the method and thermal treatment used, it is possible to obtain different tungsten phases: crystal-like hydrates, amorphous or nanocrystalline  $m\text{-WO}_3$ .
2. The final product at 400 °C is the same ( $m\text{-WO}_3$ ) irrespective of the methods applied. The average crystallite size is 15 nm.
3. With peroxo sol-gel methods, nanocrystalline  $\text{MoO}_3$  powder is obtained. The average crystallite size is 25 nm.
4. In comparison with  $\text{WO}_3$ , crystallization of  $\text{MoO}_3$  starts at a lower temperature (200 °C).
5. It is established that the use of the method B is more appropriate for the preparation of nanocrystalline  $\text{WO}_3$  and  $\text{MoO}_3$ .

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